

A Peer Revieved Open Access International Journal

www.ijiemr.org

COPY RIGHT



2021IJIEMR.Personal use of this material is permitted. Permission from IJIEMR must

be obtained for all other uses, in any current or future media, including reprinting/republishing this material for advertising or promotional purposes, creating new collective works, for resale or redistribution to servers or lists, or reuse of any copyrighted component of this work in other works. No Reprint should be done to this paper, all copy right is authenticated to Paper Authors

IJIEMR Transactions, online available on 17th Sept 2021. Link

:http://www.ijiemr.org/downloads.php?vol=Volume-10&issue=ISSUE-09

DOI: 10.48047/IJIEMR/V10/I09/28

Title Electrical and Optical characterization of Malonic acid doped Poly Vinyl Alcohol polymer electrolytes

Volume 10, Issue 09, Pages: 246-254

Paper Authors A.Alakanandana





USE THIS BARCODE TO ACCESS YOUR ONLINE PAPER

To Secure Your Paper As Per UGC Guidelines We Are Providing A Electronic Bar Code



A Peer Revieved Open Access International Journal

www.ijiemr.org

Electrical and Optical characterization of Malonic acid doped Poly Vinyl Alcohol polymer electrolytes

A. Alakanandana

Department of BS, G. Narayanamma Institute of Technology & Science, Hyderabad, Telangana, India

Corresponding author email ID: alakanandana@gmail.com

Abstract

Novel proton conducting solid polymer electrolyte membranes based on Poly vinyl alcohol (PVA) with Malonic Acid as dopant are prepared by solution cast technique with varying doping concentrations up to 40 wt.%. The electrical conductivity and optical absorption of pure PVA and Malonic aciddoped PVA electrolytes are investigated. DC conductivity behavior is studied in the range 303K to 373K. It is found that PVA: Malonic acid (70:30) electrolyte exhibits the maximum conductivity. The electrical conductivity initially increases with increasing dopant concentration and then showed a decrease beyond 30 wt.% concentration. The increase in conductivity is attributed to formation of charge transfer complexes while the decrease for concentrations above 30 wt.% is due to segregation. Optical absorption studies are made in the wavelength range 200-600 nm and the values of optical band gap (direct and indirect) are estimated. The results obtained are presented and discussed.

Keywords: Electrolyte membranes, electrical conductivity, optical absorption, Malonic acid doped PVA electrolytes, optical band gap.

1. INTRODUCTION

Now-a-days the polymer materials are of large interest because of their low cost, easy processability, low weight, high quality surfaces and easy fabrication etc. In recent years, studies on the electrical and optical properties of polymers have attracted much attention due to their applications in optical devices [1, 2]. The optical properties of polymers can be suitably modified by

addition of dopant or other polymer with the host [3].The potential application of ionically conducting polymers as solid electrolytes in a variety of electrochemical devices arises from their advantages such as high energy density, electrochemical stability and easy handling [4]. Within the realm of solid polymer electrolytes, proton conducting



A Peer Revieved Open Access International Journal

www.ijiemr.org

polymer electrolytes have received much attention because of their unique application as solid electrolyte in various electrochemical devices such as fuel cells, sensors, electrochromic displays and windows etc. [5].

The recent challenge is to develop a low cost proton conducting polymer electrolyte with high ionic conductivity, good and mechanical stabilities. dimensional Many researchers have studied the behaviour of proton conducting polymer electrolytes based on the host polymers such as poly (vinyl alcohol) (PVA), poly (ethylene oxide) (PEO), poly (ethylene imide) (PEI), poly acid) (PAA), (N-vinyl (acrylic poly pyrrolidone) (PVP) [6-13] doped with inorganic acids or ammonium salts. However, proton-conducting polymer complexes with inorganic acids suffer from degradation and chemical mechanical integrity making them unsuitable for practical applications [14]. Therefore, in this paper a novel effort is made to use of the organic acids like Malonic acid (to PVA polymer) as a dopant to prepare a proton conducting polymer and their electrical and optical characterizations are carried out.

2. EXPERIMENTAL

2.1Preparation of polymer electrolyte films

Pure PVA (Mw 1, 40,000 from AR chemicals) and PVA complexed with Malonic acid in various compositions (90:10), (80:20), (70:30) & (60:40) by wt% ratio, are prepared using the solution caste technique. PVA &Malonic acidare dissolved in double distilled water, the solutions obtained are thoroughly stirred for 10-12 hours to get homogenous mixture. The stirred

solution is caste onto polypropylene dishes and allowed to evaporate slowly at 50°C. After one or two days solid polymer layers are formed as thin membrane with nearly 100 microns thickness at the bottom of dishes. These membranes are dried in vacuum $(10^{-3}$ torr) to eliminate the residual traces of water then carefully separated from the dishes and stored in evacuated desiccators.

2.2Electrical and optical measurements

D.C. conductivity is measured using indigenously built instrument in the temperature range 300-373K with Keithley programmable electrometer (model No.196). Thickness of the films is measured by Mitutoyo thickness gauge (no.7301, range 0.01mm to 10mm). Optical absorption spectra are recorded at room temperature in the wave length 200-600 nm using optical spectrometer.

3. RESULTS AND DISCUSSION

3.1 Electrical properties

The electrical properties of the polymer electrolyte can be enhanced by modifications to the polymer host or changing the dopant salt/acid or its proportion. In addition to this, they can further be enhanced by supplying thermal energy.

3.1.1 Composition dependence of conductivity.

The influence of the nature and the concentration of the incorporate dopant on the electrolyte is complex. Essentially, polymer electrolyte is formed when the acid/salt consisting of a polarizing cation joins a large anion of delocalized charge to



A Peer Revieved Open Access International Journal

www.ijiemr.org

minimize the lattice energy. The dopant affects the conductivity through crystalline complex formation, intra-molecular crosslinking of the polymer chains and the degree of salt dissociation (the number of charge carriers it can provide). When more salt is dissolved in to the polymer matrix, the conductivity increases as a result of addition of charge carriers. As the salt concentration in a solution is increased, the inter-ionic distance decreases and ion-ion interactions become progressively more significantwhich contribute for enhancement in conductivity.

It is known that in semi-crystalline polymers, the dopant forms Charge Transfer Complex (CTC) and is expected to reduce the barrier between the trapping sites. This reduction in the barrier provides a conduction path through the amorphous regions of the polymer matrix. The decrease in the activation energy and increase in the mobility of carriers results in an enhancement of its conductivity [15, 16].

However, dopant concentration as increased above a certain limit. the conductivity reaches a maximum and then falls. It has been deduced that at high dopant formation immobile concentrations. of aggregated regions will contribute to the fall in conductivity (Gray, 1991).

The variation of dc conductivity (σ) as a function of dopant (acid) content by wt% in Poly Vinyl Alcohol (PVA) and Malonic acid at various temperatures is as shown in Fig.1. The increase in the ionic conductivity with the increase in the composition of the dopant is primarily attributed to increased proton carrier concentration. The movement of protons through the polymer matrix is

assisted further by the large amplitude of the polymer segmental motion. The segmental mobility of polymer chains is much higher in amorphous regions than in crystalline.

Addition of more acid in to the polymer not only contributes more protons for conduction but also increases the amorphous regions and at the same time reduces the crystallinity. The continuous increase in the conductivity of polymer electrolytes with increasing salt/acid concentration is thus attributed to decrease in the degree of crystallinity. Therefore, conductivity increases as the degree of crystallinity decreases or in other words, as the flexibility of the polymeric backbone increases.



Fig.1 Composition vs conductivity of PVA: Malonic acid polymer electrolytes with temperature as parameter.

Jeon et al (17) reported that, the conductivity of the polymer electrolytes increased with increasing salt concentration, but further addition of the salt caused the conductivity to decrease. At higher



A Peer Revieved Open Access International Journal

www.ijiemr.org

concentrations, the conductivity decreases because of the increasing influence of ion pairs, ion triplets and higher ion aggregations which reduce overall mobility and number of effective charge carriers. This effect is observed for the cases of dopant acid concentration in the region above 30 wt% of salt (acid) in the PVA polymer.

3.1.2 Temperature dependence of conductivity

In general, the temperature dependent conductivity (σ) of polymers may be expressed as in Eqn.1

$$\sigma = (\sigma_0/T) \exp(E_a/kT) \qquad ---- (1)$$

where σ_0 is the pre-exponential factor, E_a the activation energy, k the Boltzmann constant and T is the absolute temperature.

The increase in the conductivity with temperature is also partially due to the transition from crystalline/ semi-crystalline phase to amorphous phase.

According to the Vacancy diffusion theory, for the protons to jump from one lattice site to other lattice site, need energy to break bonds with neighbors and to cause the necessary lattice distortions during jump. This energy comes from the thermal energy of atomic vibrations ($E_{av} \sim k_B T$). Due to the thermal energy supplied, a small free volume is created around segments. At low temperatures Arrhenius model was assumed cation transport where the and its contribution to conductivity is a result of cation jump into neighbouringvacant sites and hence, increases the proton conductivity to a higher value [18, 19]. This hopping

mechanism between coordinating sites is essentially in the region below T_g .

However, above Tg, no dependency on comparable temperature to Arrheniusbehaviour could be obtained. Poly Vinyl Alcohol is a linear polymer with carbon chain as the back bone, the polymer chains which are less entangled are capable of causing electrical conductivity. At temperatures beyond the glass transition T_g, the increase in temperature the conductivity with temperature is mainly due to hopping mechanism between coordination local structural sites. relaxation and segmental motion of polymer [20,21]. As the amorphous region increases, the polymer chain acquires faster internal modes in which bond rotations produce segmental motion. This, in turn, favours the hopping inter-chain and intra chain proton movements and the conductivity of the polymer thus becomes appreciably high [22]. Thus it suggests that, the result may be effectively represented by the non-linear dependence and follows the Williams-Landel-Ferry (WLF) mechanism This mechanism can be better [23]. described by Vogel-Tammann-Fulcher (VTF) models and VTF equation.

At the significantly high temperatures (beyond T_g), the polymer can expand easily and produce more free volume which is sufficient to move into the free volume not only for protons but also for solvated molecules or even polymer segments [24]. Therefore, the DC conductivity (log (σ)) versus reciprocal of temperature can be interpreted in terms of the empirical VTF equation. Thus, the result may be effectively represented by Vogel–Tammann–Fulcher (VTF) models and more appropriate to this



A Peer Revieved Open Access International Journal

www.ijiemr.org

situation that the behaviour is explained using the Vogel–Tamman–Fulcher (VTF) expression of Eqn.2 corresponding to this region, based on the free volume concept (Kim *et al* 1996).

$$\sigma = AT^{-1/2} \exp(-B/T - T_0) \qquad - - - -$$
(2)

where A and B are constants and T_0 is the reference temperature. Constant A is related to the number of charge carriers in the electrolyte and B is related to the activation energy of ion transport associated with the configurational entropy of the polymer chains.

The variation of dc conductivity as a function of inverse temperature for pure PVA and composite polymer electrolytes with different compositions of PVA: Malonic acid is shown in Fig.2 in the temperature range of 298-373K. From the plots the proton conductivity is found to increase with the increase of temperature in pure PVA as well as in all the composite (complexed) polymer electrolytes.



Fig.2 Conductivity $log(\sigma)$ vs 1/T plots for PVA:Malonic acid composite polymers

This is in accordance with the expected behaviour from all materials except metals. From the plots it is observed that at low temperatures, Arrhenius behaviour is exhibited and at high temperatures, VTF (Vogel-Tamman-Fulcher) behaviour is exhibited. Both the regions correspondingly have two distinct activation energies. А number of other polymer electrolyte systems including those based on **PVA** [25,26,27,28,29] have also exhibited similar behaviour.

In order to further analyze the conduction mechanism, the conductivity values are evaluated for the regions below and above the glass transition temperature T_g using the conductivity plots shown in fig.2. These values are tabulated in the table.1.

Table 1: Conductivity values and Activation
Energy values of PVA doped Malonic acid based
polymer electrolytes

Dahuman alaatus kuta	σ in S/cm	σ in S/cm	AE in eV	AE in eV
rolymer electrolyte	at 303K	303K at 373K A		At>Tg
Pure PVA	5.62x10-10	6.15 x10-9	0.13	0.61
PVA: Malonic acid (90:10)	7.33x10-8	3.73x10-7	0.23	0.44
PVA: Malonicacid (80:20)	2.56x10-7	4.796x10-7	0.16	0.08
PVA: Malonic acid (70:30)	1.17x10-6	1.23x10-5	0.24	0.35
PVA: Malonicacid (60:40)	2.42x10-7	1.04x10-6	0.37	0.04

From the obtained values of activation energies, it is inferred that both the regions do not exhibit the same behaviour of conductivity as the values are distinct for the two regions and the pattern of variation of these values, with varying dopant



A Peer Revieved Open Access International Journal

concentration, is also different for the region below T_g from the one corresponding to the region above T_g .

3.2 Optical properties

The information about band structure can be studied by an optical absorption. In indirect bandgap semiconductors, the transition of electron from conduction band to valence band is always associated with a phonon of the appropriate magnitude. Davis and Shalliday [30] reported that in polymer electrolytes, near fundamental band edge both direct and indirect transitions occur and can be observed by plotting $\alpha^{1/2}$ and α^2 as a function of energy hv. The relationship in eqn.3 and eqn.4 are based on the analysis of Thutupalli and Tomlin [31]

$$(hvan)^2 = C_1 (hv - E_{gd}) ---- (3)$$

 $(hvan)^{1/2} = C_2 (hv - E_{gi}) ---- (4)$

for direct and indirect band semiconductors respectively, where hv is the photon energy, E_{gd} is the direct band gap, E_{gi} is the indirect band gap, n is the refractive index, α is the absorption coefficient and C_1, C_2 are constants.

The absorption coefficient (α) is calculated from the absorbance (A). The variation in absorbance with wavelength of the incident radiation is shown in fig.3. The absorbance peak has shifted in the case of doped PVA suggesting the occurrence of complexation. After the necessary correction for reflection, the absorption coefficient (α) is determined using the relations in eqn.5 and eqn.6.

 $I = I_0 \exp(-\alpha x) \qquad \qquad ---- (5)$ Hence

$$\alpha = \frac{2.303}{x} \log\left(\frac{I}{I_o}\right) = \left(\frac{2.303}{x}\right) A^{----} (6)$$

www.ijiemr.org

where x is the thickness of the sample.



Fig.3Optical absorption spectra of pure and Malonic acid doped PVA polymer electrolytes

To determine the width of the band gap $(\alpha h\nu)^2$ is plotted as a function of photon energy $(h\nu)$ and shown in fig.4. The direct band gap value is determined by extrapolating the linear portion of the curve on to the energy axis. The intercept on the x-axis gives the direct band gap energy value.





A Peer Revieved Open Access International Journal

Fig.4(α hv)² vs hvplots of pure and Malonic acid doped PVA polymer electrolytes

When a direct band gap exists, the absorption coefficient has the following dependence, shown in eqn.7, on the energy of incident photon [32, 33]

$$h\nu\alpha = C (h\nu - E_g)^{\frac{1}{2}}$$
(7)

where E_g is the band gap, $C(=4\pi\sigma_0/nc\Delta E)$ is a constant dependent on the specimen structure, α is the absorption coefficient, ν is the frequency of the incident light and h is the Planck's constant. The curves of $(\alpha h\nu)^2$ vs. hv are plotted and the direct band gap values are obtained which are tabulated in the Table2.

For indirect transitions, which require phonon assistance, the absorption coefficient has the following dependency, shown in eqn.8, on photon energy [32, 33].

$$hνα = A (hν- E_g + E_p)^2 + B (hν- E_g - E_p)^2$$

where E_p is the energy of the photon associated with the transition and A and B are constants depending on the band structure. Again, the indirect band gap values are obtained from the $(\alpha h\nu)^{1/2}$ vs hv plot and is shown in fig.5.



www.ijiemr.org

Fig.5. (αhv)^{1/2}vs hv plots of pure and Malonic acid doped PVA polymer electrolytes

Following the same procedure adopted for determining the direct band gap energy, the indirect band gap value is also determined by extrapolating the linear portion of the curve on to the energy axis. The intercept on the xaxis gives the indirect band gap energy value. These values are also tabulated in the table 2.

From the significant values obtained for indirect band gap energies for various polymer electrolyte samples clearly indicate that the samples under consideration exhibit not only direct band gap transitions but also the indirect band gap transitions. It is clear from the table, that the direct band gap and indirect band gap values displayed a decreasing trend on complexing with dopant acid up to a dopant concentration of 30%wt. This type of behaviour was also observed in many other polymer electrolytes [34-37].

The decrease in optical band gap energy on doping, up to a dopant concentration 30 wt.%, may be explained on the basis of the fact that the incorporation of small amounts of dopant forms charge transfer complexes (CTCs) in the host lattice [38]. These charge



A Peer Revieved Open Access International Journal

www.ijiemr.org

transfer complexes increase the electrical conductivity by providing additional charges in the lattice. This results in a decrease of optical band gap energy.

Table 2.Optical band gap values for PVAcomplexed with Succinic acid polymerelectrolytes.

Polymer electrolyte	Direct bandgap energy (eV)	Indirect bandgap energy (eV)
Pure PVA	5.65	5.20
PVA:Malonic acid (90:10)	5.32	4.93
PVA:Malonic acid (80:20)	5.28	4.88
PVA:Malonic acid (70:30)	5.23	4.73
PVA:Malonicacid (60:40)	5.26	4.76

When the dopant concentration increases further, it leads to segregation of the dopant in the host matrix. These molecular aggregates impede the motion of charge carriers resulting in decreased conductivity and hence increased optical band gap energy.

4. CONCLUSIONS

The conductivity of pure PVA is enhanced by doping with Malonic acid, a member of dicarboxylic acids and one of the organic acids, due to additional protons supplied by the dopant.

It is found that the maximum conductivity and minimum bandgap energy values, for both direct and indirect transitions, are obtained from the complexed PVA with Malonic acid when in 30% wt. proportion.

The DC conductivity of 1.23×10^{-5} S/cm at 373K is an encouraging value which

suggests that a complexed solid polymer electrolyte membrane made from PVA with Malonic acid as dopant is a promising candidate for use in an electrochemical cell.

REFERENCES

- 1. Peppas NA, Bures P, Leobandung W, Ichikawa H (2000) Eur J Pharm Biopharm 50:27–46.
- Osada Y, Gong JP, Tanaka Y (2004) Polymer gels. J MacromolSci Part C Polym Rev 44(1):87–112.
- A Rawat, H K Mahavar, S Chauhan, A Tanwar, P J Singh(2012) Indian J Pure & ApplPhys, Vol 50: 100-104.
- 4. Takada N, Koyama T, Suzuki M, Kimura M, Hanabusa K, Shirai H, Miyata S (2002) Polymer 43:2031
- 5. Zukowska G, Wieczorek W, Kedzierski M, Florjanczyk Z Solid State Ionics (2001) 144:163
- 6. Stainer M, Hardy LC, Whitmore DH, Shriver DF (1984) J. Electrochem. Soc 131:784
- Daniel MF, Desbat B, Cruege F, Trinquit O, Lassegues J.C (1988) Solid State Ionics 28– 30:637
- 8. Daniel M.F, DesbatB,Lassegues JC (1988) Solid State Ionics 28–30:632
- 9. Hashmi SA, Kumar A, Maurya KK, Chandra (1990) S J Phys D: Appl. Phys. 23:1307
- 10. Chandra S, Hashmi SA, Prasad G (1990) Solid State Ionics 40:6541
- Maurya KK, Srivastava N, Hashmi SA, Chandra S (1992) J. Mater. Sci. 27:6357
- 12. K.K. Maurya, S.A. Hashmi, S. Chandra, J. Phys. Soc. Jpn. 61 (1992) 1709.
- Donoso P, Gorecki W, Berthier C, Defedini F, Poinsignon C, Armand MB (1988) Solid State Ionics 28–30:969
- Rikukawa M, Sanui K (2000)Prog. Polym. Sci. 25:1463
- 15. A.B.Mansour, J.Appl.Polym.Sci. 91,3167 (2004)



A Peer Revieved Open Access International Journal

www.ijiemr.org

- 16. B.E.Kilbride,J.N.Coleman,J.Fraysse,P.Fournet,M .Cadek,Drury,S.Hutzler,W.J.Roth Blau, J. Appl. Phys. 92, 4024 (2002)
- L. M. Carvalho, P. Gueágan, H. Cheradame, A. S.Gomes, European Polymer Journal 36 (2000) 401-409
- L. Othman, K. W. Chew, Z.Osman, Ionics 13 (2007) 337–342
- 20. H.Devendrappa, U.V. Subba Rao and M.V.N. Amibika Prasad, J. Power Sources 155 (2) (2006) 368.
- 21. H. Vogel, Phys. Z. 22 (1922) 645.
- 22. V.V.R.Narasimha Rao, T.Mahendar, B.Subba Rao, J.Non-cryst.Solid 104 (1) (1988) 224.
- 23. M.L. Williams, R.F. Landel, J.D. Ferry, J. Am. Chem. Soc. 77 (1995) 3701.
- 24. T. Uma, T. Mahalingam, U. Stimming, Materials Chemistry and Physics 90 (2005) 245–249
- 25. Ch. V. Subba Reddy, A.K.Sharma, V.V.R.Narasimha Rao, J Polymer 47 (2006)
- 26. AmrithaBhide, K. Hariharan, J Power Sources 159 (2) (2006) 1450.
- 34. V. BalasubramanyamAchari, T.Janakirami Reddy,A.K.Sharma, R.Narasimha Rao,(2007).J.Ionics

- Jeon, J. D.; Kwak S. Y.; Cho, B. W. J. Electrochem. Soc. 2005, 152, A1583.
- 27. S.K.Patel, R.B.Patel, A.Awadhia, N. Chand and S.L.Agarwal. PRAMANA –Journal of Physics, vol.9,No.3(2007) 467-475.
- Navin Chand, Neelesh Rai, S L Agarwal and S K Patel. Bull. Mater. Sci., Vol. 34, No. 7, December 2011, pp. 1297–1304.
 Indian Academy of Sciences
- G. Hiran Kumar, S.Selvasekarapandian, N. Kuwata, T. Hattori. J Power Sources 144 (2005) 262.
- 30. Davis and D.S. and Shalliday, T.S. Phys.Rew.118,1020(1960)
- 31. Thutupalli , G.M. and Tomlin, J. Phys. D:Appl. Phys.9.1639(1976
- V.Raja, A.K.Sharma, V.V.R.Narasimha Rao. J. Matter Lett. 57 (2003)4678
- V.M.Mohan, V.Raja,P.B. Bhargav, A.K.Sharma,V.V.R.Narasimha Rao. J Polym Res 14 (2007)283.
- 37. V.M.Mohan, V.Raja,P.B. Bhargav, A.K.Sharma,V.V.R.Narasimha Rao, J Polym Res
- 35. C. Uma Devi, A.K.Sharma, V.V.R.Narasimha Rao, Matter Ldtt. (20(20)26)3167.
- V.Raja, A.K.Sharma, V.V.R.Narasimha Rao, J. Matter Lett. 57 (2003)4678.
- Ch. Ramu, Y.R.V. Naidu, A.K. Sharma, Ferroelectrics 159 (1994) 275.